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X-RAY PHOTOELECTRON STUDY OF Yb-DOPED InP

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X-ray photoelectron spectra of core levels are reported for InP:Yb. Crystalline InP, doped with Yb to a level of 0.5 at.%, was grown by the synthesized solute diffusion method. An analysis of the core-level spectra of the constituent components, i.e. In $3d_{5/2}$ and P 2p, revealed a minor influence of the surface oxide species, mainly in the phosphate-like form. The spectrum of the Yb 4d core level was also recorded. The energy of the Yb $4d_{3/2}$ peak was found identical to that in Yb metal, whereas the $4d_{5/2}$ peak was found to be shifted to higher binding energies. This effect was found comparable to the case of advanced oxidation of Yb thus confirming its high reactivity, even as a bulk dopant. The data give also a rare experimental example of detection of bulk dopant atoms in a semiconductor matrix by X-ray photoelectron spectroscopy at the limit of detectability.

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1. Introduction

In the decade of the eighties, indium phosphide has proved to be a very promising semiconductor material for various applications owing to its intrinsic properties, such as high electron mobility and high radiation tolerance. This obviously implied the interest in its potential applications, such as high-speed electronic devices, high-performance space solar cells and integrated optoelectronic devices. In the latter case, the interest has been additionally enhanced after successful incorporation of Yb into InP by bulk [1] or epitaxial growth [2], which led to the observation of the Yb⁺³-related electroluminescence in the emission band around 1.24 eV [3, 2].

The perspective of electronic device applications has been also stimulating for studies of InP surface preparation with the use of X-ray photoelectron spectroscopy (XPS), e.g. [4-6]. In the case of Yb-doped InP, there is a lack of reports concerning XPS investigations (see Ref. [7]). Hence, the goal of the present paper was to provide preliminary results of an XPS characterization of InP:Yb.

2. Experiment

Bulk InP crystals were grown by the synthesized solute diffusion (SSD) method [8] at the Institute of Physics, Polish Academy of Sciences. 0.5 at.% of Yb metal was added to the starting material to grow an Yb-doped crystal. The growth temperature was 850° C. The obtained crystalline ingot was 4 cm long with a diameter of 1 cm. The sample (1 mm thick) prepared for this experiment was cut from the middle part of the ingot, perpendicularly to the main axis. Its surface was subjected to fine polishing and was finally etched in a 1% Br₂-methanol solution. Electron microprobe (JEOL JSM-50A) analysis enabled us to estimate that the real Yb concentration in the sample remains below its assumed value (0.5 at.%).

Core-level X-ray photoelectron spectra were measured at room temperature in a vacuum generator (VG) Escalab 210 spectrometer using an Al K_{α} (1486.6 eV) irradiation source. The photoelectrons were detected normal to the surface.

3. Results and concluding remarks

Core-level XPS spectra of the constituent elements of InP, namely P 2p and In $3d_{5/2}$, are shown in Figs. 1 and 2a, respectively.



Fig. 1. Photoelectron spectrum of the P 2p level excited with Al K_{α} radiation from the InP surface.

In particular, the P 2p spectrum (Fig. 1) is readily observed as a doublet. One can distinguish there the main peak at 128.7 eV ascribed to P in InP (bulk component), and the minor peak around 133.6 eV (surface component), which is indicative of phosphorus bound in phosphates, introduced by surface oxidation (see [4, 5, 7]).

The photoelectron spectrum of the In $3d_{5/2}$ core level (Fig. 2a) exhibits an asymmetric shape, characteristic for an overlap of bulk- and surface-component peaks [4, 5]. Its numerical analysis, performed by fitting with Gaussian-Lorentzian peaks, enabled us to distinguish the dominant bulk-related peak ($E_{\rm B} = 444.2 \text{ eV}$) from the weaker surface-related contribution ($E_{\rm B} = 445.1 \text{ eV}$).

A similar analysis was also performed for the O 1s photoelectron spectrum, where four constituent subpeaks were found (Fig. 2b).



Fig. 2. Photoelectron spectra of In $3d_{5/2}$ (a) and O 1s core (b) levels in InP and results of their numerical analysis (for details see text and Table).

TABLE

Results of a numerical analysis of the In $3d_{5/2}$ and O 1s core-level photoelectron spectra measured from the InP:Yb surface. FWHM denotes the full line width at half-maximum.

Core level	Peak No.	$E_{\mathbf{B}}$ [eV]	FWHM [eV]	Area [%]
In 3d _{5/2}	1	444.2	1.38	55
	2	445.1	1.96	45
O 1s	1	530.0	2.35	13
	2	531.8	2.29	64
	3	533.2	2.16	17
	4	534.5	1.83	6

Table summarizes the results of the numerical analysis of both the In $3d_{5/2}$ and O 1s XPS peaks, retaining the same notation of subpeaks as in the respective figures.

Taking into account the above evidence (Figs. 1 and 2, Table) and the literature data [5] one can conclude that in a surface- and subsurface region of InP, accessible to XPS, there is a noticeable (although minor) influence of different native oxide species.

In particular, the peak $E_1(O \ 1s)$ is due to the presence of indium oxide (In_2O_3) [5]. The parameters (energy, FWHM) of the peak 2 from the O 1s spectrum indicate that it probably results from superposition of the contributions given by $In(OH)_3$, $In(OH)_3 \cdot xH_2O$, In_2O_3 as well as the phosphate species like $InPO_4$ and $InPO_4 \cdot xH_2 O$ [5]. The $E_3(O \ 1s)$ peak (Table) can be attributed to the existence of $In(PO_3)_3$, that also contributes to the $E_2(O \ 1s)$ peak [5]. The weak $E_4(O \ 1s)$ peak also points at the residual admixture of $In(PO_3)_4$ [5].

The peak 1 of the In $3d_{5/2}$ spectrum (Table) has relatively low FWHM (1.38 eV), that is typical for unperturbed bulk-related component [5]. On the other hand, broadening of the subpeak $E_2(\ln 3d_{5/2})$ (FWHM = 1.96 eV — Table) is indicative of the overlapping contributions from different (above-mentioned)

oxides and phosphate-like species (see [5]). The split of E_2 from the E_1 peak is due to relatively higher (compared to phosphorus) electronegativity of oxygen [9], bound to In in the oxide or phosphate configuration.

The apparent existence of native oxides in the studied case is not surprising, since we intentionally avoided a surface-deoxidation etch in order to check the influence of partial surface oxidation on the binding of Yb in the InP lattice.



Fig. 3. Photoelectron spectrum of Yb 4d levels taken from the InP:Yb surface.

The XPS spectrum of the Yb 4d core-level is presented in Fig. 3. One can distinguish there the most pronounced peak (denoted by A) centered at 186.2 eV and a weaker one (B) at 190.6 eV. There is also an additional broad feature (C) located around 199 eV. A comparison with earlier data [10, 7] enabled us to assign the peaks A and B with the spin-orbit split 4d levels $(4d_{5/2} \text{ and } 4d_{3/2})$, respectively). The energy of the $4d_{5/2}$ peak, $E_{\rm B}(A)$, (together with its intensity ratio vs. $4d_{3/2}$ peak) is a specially useful indicator in monitoring the oxidation state [7, 10] and, in general, the valency state. The value of $E_{\rm B}({\rm A})$ is found here to be shifted (from that of Yb metal) to higher binding energy, which coincides well with that of Yb_2O_3 [7]. On the other hand, the energy of the peak B (190.6 eV) agrees very well with that of the corresponding $4d_{3/2}$ peak of Yb metal [10, 7] and a partially oxidized Yb [10], whereas it slightly differs from that of pure Yb_2O_3 $(E_{\rm B} \cong 192 \, {\rm eV} \, [10, 11])$. Despite of its complexity, the spectrum considered (Fig. 3) seems indicative of the advanced oxidation stage of Yb, with dominant +3 valency state, characteristic of Yb₂O₃ [7, 10]. This can be also supported by an appearance of the feature C, which is usually attributed to the $4f^{13}$ configuration of Yb ion (see also [12]). This enables us to judge that in our case of partially oxidized surfaceand subsurface region of InP crystal the Yb dopant ions are mainly bound to oxygen and form Yb₂O₃ clusters in the host matrix, thus confirming a very high reactivity of Yb. The earlier EPR studies [13] performed for bulk InP:Yb crystals, grown by the same method, have proved that in the absence of oxygen an Yb^{3+} ion (electrically neutral with respect to In³⁺) occupies an In site in the lattice. Therefore, taking into account the oxygen concentration gradient in a subsurface region, one can not exclude here a certain influence of these latter Yb ions on

the Yb 4d XPS spectrum (Fig. 3), especially that the photoelectron signal was recorded in the maximum bulk sensitivity mode. This could be obviously verified when repeating the measurement for the clean surface of the same crystal.

Our data represent the first, although preliminary, XPS analysis of Yb impurities in InP. They are also a rare experimental example of the XPS detection of bulk dopant atoms in a semiconductor matrix in the doping range of ≤ 0.5 at.%, i.e. at the XPS detectability limit [14].

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